H Shahan Department of Chemical Engineering College of Engineering and Petroleum University of Kuwait
P.O. Box 5969 13060 Safat, Kuwait

ABSTRACT

Causes of failure of a vacuum residue steam generator tubes were investigated using visual, microscopic and chemical methods. It was found that the
main cause of such failure was due to stress corrosion cracking. Suggestions were made to control causes of such failure.

RESUMEN

Se investigó las causas de la falla de los tubos de un generador de vapor al vacío utilizando cos. Se encontro que la causa principal de tales fallas fue debida a corrosión por tensiones. hacen sugerencias para controlar las causas de tal falla.

INTRODUCTION

Stress corrosion cracking (SCC) of steels \mathbf{m} corrosive media (1) can occur when such steels have conside stresses. These tensile stresses may range
from as low as 10% of the yield stress to as high
as 70% the yield stress for the SCC to occur. For each alloy-environment combination, there is an ef fective of threshold stress. These tensile stres-
ses may be due to any cause : applied, residual, thermal or welding.

Corrosion products can cause stress of up to 10000 psi in constricted regions. The presence of
chlorides with dissolved oxygen or any other oxidiof zing agents in a given aqueous environment can cau-
se SCC to austenitic stainless steels. Absence of se such oxidizing agents or oxigen will inhibit
stress corrosion cracking. SCC is accelerated with temperature like most chemical reactions.

Metallurgical factors such as alloy chemical Metallurgical factors such as alloy chemical
composition, grain orientation, presence of dislo-
cations, etc. are important factors for the SCC
to occur. The mechanism by which SCC occurs is not
well understood. In general

-41

Rev. Téc. Ing., Univ. Zulia, Vol. 10, No. 1, Edición Especial, 1987 10° Aniversario de la Revista Técnica de la Facultad de Ingeniería

STRESS CORROSION CRACKING OF TUBE OF A VACCUM RESIDUE STEAM GENERATOR: A CASE **STUDY**

creases tremendously as the radius of the potch decreases. Pardue et al. (2) using audio-amplifica-
tion methods showed that a mechnical step (i, e) . pings) can occur and were heard during crack propagation.

Methods for controlling SCC are one or more
of the following : lowering the stress below the
threshold value if one exists, eliminating the
critical environment by degasification, pH adjust-
ment or any other means, changi thodic protection, and adding inhibitors if po
ble such as phosphates or any other inorganic possi or organic corrosion inhibitors.

HISTORY OF THE PROBLEM

 $\begin{array}{c} \end{array}$

A number of tube failures occurred in two steam generators after a short period of service. Most of the failures were in the hot zones and to-
wards the outside of the bundles. All the removed tubes showed transverse fracture which was located call adjacent to the expanded section of the tubes as
shown in figure (1). The right hand end of the
tube is the section that was expanded into the tube tube is the section that was expansive the cover
sheet hole by a process of roller expansion over the first 2 inch of the tube length. The outer
surface of this section is fairly bright. The frac-
ture is adjacent to this section. The remainder of
the tube shown in the figure extends into the main body of the generator.

Material specification for the tubes is ASTM Al79 which is a cold drawn condenser tube material
having the composition : 012% carbon, 0.35% manganese, 0.05% sulfur and 0.05% phosphorous.

This kettle type steam generator is one $\alpha \in$ seven similar units using different plant products (e.g. vacuum residue, waxy distallation, etc.)
producing steam to be used for the different for plant processes (e.g. vacuum distillation, steam tracing processes (e.g. vacuum distillation, steam tracing
of piping network, heating colls in storage tanks,
etc.). The plant which is a newly operated bitumen
plant is located inside a petroleum refinery com-
plex in Gulf Corpor

The steam generator under investigation was manufactured from carbon steel. The tube side fluid
was vacuum residue containing 5.2% sulfur and some was vacuum residue containing 5.2% sulfur and H₂S at high temperature. The agressive nature of this fluid resulted in a failure of two steam generators working with this fluid. The failure was due to leakage of the tube side fluid (vacuum residue or Bitumen) to the shell side fluid as shown by water and steam analysis of these generators. Unplanned shutdown of the whole plant took place to investigate and rectify che cause of faiIure.SampIes of the cracked tubes were puIIed out for investigarion and were renewed usiog tubes of the same material (ASTM Al79).

Failure analysis showed that the reason of
uge was stress corrosion cracking SCC of the Ieakage was stress corrosion cracking SCC of the tubes which have been overexpanded during manufacture and enhanced by the improper water treatment (high pH). Appendix shows the conditions and design data for the steam generator and its fluids.

EXPERIMENTAL

The outer and inner surfaces of the tubes were examined at different positions using low power optical microscope. An overall vlew of the failed tubes were taken by a camera with a color film. (Figu-
res a, b, c, d and e) Samples were taken from different areas of the tube, prepared metallographically and examined under the optical microscope. Wall thickness of the tube was measured in a number of areas ueing mlCTometer .

RESULTS

1. THE OUTER SURFACE:

The outer suface (excluding the expanded portion of the tube) was covered with a fairly smooth black scale. However, there were localised regions black scale, However, there were localised regions
of corrosion which were visible at low magnifica-
biggs (Figure 2) On removal of this scale it was tions (Figure 2). On removal of this scale it was
found that the surface was fairly deeply pitted found that the surface was fairly deeply pitted (Figure 2) on the expanded side of the fracture, but no pitting was visible on the expanded section of the tube. Instead there was found to be a large number of longitudinal creacks on the surface of the expanded part of the tube (Figure 3).

A more detailed examination of the outer surface was carried out on a cross section of the tube. Figure 6 shows the outer surface in the expanded portion of the tube; the surface is quite smooth and free of pits. On the other side of the fracture face this surface showed deep pits with cracks beginning to spread from some of them; one such craek is shown in Figure 4.

On a part of the outer surface remote from the fracture there was found to be a region of localised deformation and associated cracks (Figure 5).

2. THE INNER SURFACE:

The general appearance of the inner surface
ne of fairly loose corrosion products which was one of fairly loose corrosion products which were bright rust coloured at, what was considered were bright rust coloured at, what was considered
to be, the bottom of the tube. At high magnificato be, the bottom of the tube. At high magnifica-
tions (Figure 6) it can be seen that there is general corrosion and this is also appparent on the cross aeetion of the tube shownin Figure 6. In addition, deep score marks were also evident over the first 4" or so of the tube (measured from the expanded end of the tube). Some of these marks are shown in Figure 6 and the result of this scoring on the deformation of the grain structure can be seen in Figure 5.

3. THE. FRACTURE SURFACE:

The whole of the fracture surface was coated by
form black scale, (in all appearances the same a uniform black scale, (in all appearances the same
as the scale on the outside of the tube). The fracas the scale on the outside of the tube). The ture was approximately transverse with a small discontinuity on the surface. Examination of this surface microscopically showed that intergranular attack had taken place and this was seen under the optical microscope (Figure 4). The fracture path follows the grain boundaries and inter granular attack can also be Seen adjacent to the fracture. Ihe intergranular regions close to the fracture surface
appear to be filled with oxides. It can also be sean from this photograph that the general microstructure of the seeel consists of large ferrite areas interspersed with regions of spheroidal cementi $t e$.

4. REDUCTION IN WALL THICKNESS:

The wall thickness of the tube was measured at a number of places along the tube and at any one section at different circumferential positions. It was found that the wall thickness in the vicinity of the fracture varied from 0.110" to 0.130" and away from the fracture from 0.118 " to 0.126 ". The lower figures appear to coincide with the tube.

DISCUSSION

It is considered that the cause of failure of the tube ls stress corrosion cracking; the following factors are thought to be most significant :

(a) Material specification : Steels containing 0.1% carbon are most susceptible to stress corrosion cracking under certain conditions. The steel used for manufacturing the tubes contains 0.12% carbon.

 $-42 -$

(b) Temperture : Stress corrosion cracking is more likely at high temperatures. Most of the tube failures have occurred in the upper section of the bundles.

(c) Stress : The level of internal stress at the junction between the expanded and non-expanded sec-
tion of the tube is probably sufficient to initiate the stress corrosion cracking.

(d) The pH of the water : High pH values in tha water will promote stress corrosion cracking, any-
thing above 8 may be considered to be high.

(e) Alkalinity : Some concentration of alkali may. occur in the small space between the tube sheet
and the unexpanded tube. This may add to the corrosheet sion problem.

(f) Break in passive film : Local corrosion may occur at a break in a passive film. The section of
the tube which has not been expanded is covered
with a uniform oxide layer, but the expanded sec-
tion is not uniformly coated, there is, therefore,
a break in the passive fracture. This may aid stress corrosion cracking.

It can be seen that all the factors that promote stress corrosion cracking are present in this ap-
plication; rapid failures are to be expected.

The route to failure is proposed as follows :

Appreciable internal stress is introduced into the tube during the expansion process, evidence for this is:

(1) Score marks on the inside surface (Fig. 6) Grain deformation on the inner surface (111) Cracking on the outer surface

In addition to this stress there will probably be applied stresses due to thermal expansion and con-
traction of the tube. The buckling of the outer tu- con traction of the twee, the buckling of the outer the corrections any also contribute to the total stress. The
corrective medium, (high temperature water of high
pH) produces pitting in the outer surface of the
tube and wher region where the final fracture occured is suffi-
cient to produce very rapid fracture of the tube,
Indirect evidence that the material is sensitive to stress corrosion can be seen in Figure 5,
some sort of deformation has occurred on the where outside of the tube away from the fracture. Here cor-
rosion and subsequent cracking has taken place over the very localised area. Although extensive corrosion has taken place from the inside of the utube, shown in the corrosion products and loss of wall thinckness, the conditions are not favourable for stress corrosion cracking to initiate at this surface.

CONCLUSIONS

STATISTICS

Failure occured by stress corrosion cracking SCC. The crack started at the outside of the Butube and spread between the grains very rapidly. The conditions leading to this failure are high temperatures, presence of internal and applied stresses,
high pH values of the water, and low carbon con-
tent of the steel. Some internal corrosion of
the tube has occured but this has almost certainly not contributed to the failure.

RECOMMENDATIONS

- 1. The pH value of the water must be reduced or shifted to a new position. It is probably
practicable to assemble the tubes without exexpansion rolling. However, it should be possible to
expand for the full thickness of the tube sheet. to. In this way the critical internal stress region
will not be within the sheet hole where concentration of corrosion products may take place.
- 2. Some assessment of present roller expanding practice may be worthwhile to determine where or not variables in fabrication procedures expansion whether may by introducing more internal stress into some tube than others.
- 3. Consider using alternative steels such as ASTMA 199. These are chromiun, molybdenum condenser tube steels which may give much better service.
There are eight possible steels quoted and which should be chosen depends very much on operating
conditions. The choice could only be made after some controlled experimentation.
- 4. It is to be expected that other tube failures
may occur at later stages in the lower tempera-
ture regions so that careful monitoring should continue.

ACTIONS TAKEN:

pH controlled to be $"8"$ New tube bundle purchased to include a higher corrosion restating material.
New Tubes Material : ASTM - A 199 Gr. T9

Expansion of tubes was made to the full length inside the tubesheet.
Also tube ends have been seal welded.

$-43 -$

b) THE RUPTURED OF THE TUBE

a) THE FAILED TUBE - OVERALL VIEW

c) THE CRACK PROPAGATION AREA

Rev. Téc. Ing., Univ. Zulia Vol. 10, No. 1, Edición Especial, 1987

Ë

d) THE TUBE THINNING

.1 SCALES ON THE OUTER SURFACES

Figure (1)

General view of tube showing expanded section, fracture and uniform cracking on the outer surface of tube in contact with water.

 $-45 -$

Rev. Téc. Ing., Univ. Zulia, Vol. 10, No. 1, Edición Especial, 1987

Ï

Figure (2)

Pittings revealed after removal of seaIe on outer surface close to the racture.

Alp is

Figure (3)

Cross-section through the outer surface af expanded portian af the tube showing pittings and cracking . .. (X = 200)

Figure (4)

Cross-seetian af fracture showing intergranular path with some branchiog into adjacent grains.

Note: the oxide scale with in the cracked boundaries... $(X = 500)$

 $-46 -$

Figure (5)

Deformed region of the outer surface
remote from the fracture showing some
cracking $(X = 200)$

Figure (6) Scoring on the inner surface.

APPENDIX

Sp.
Visc

Visc Sulf Pour

Thermal Design Data of the Steam Generator :

Shell side :

Tube side :

Flow rate of vacuum residue = $108,896 \text{ lb/hr}$

Fouling factor = $0.01 \text{ hr ft}^2 \text{ }^{\circ}\text{F/Btu}$

Inlet temperature = 650°F

Specific gravity = $API = 7.8$

Specific heat = $0.627 \text{ Btu/lb} \text{ }^{\circ}\text{F}$

Outlet temperatur

Characteristics of vacuum residue (Bitumen) :

The steam generator is of the horizontal kettle ty-
pe with overall dimensions of 58" x 67" x 197" with
2 shells and with net heat transfer area of 556
ft . It has 1768 tubes $3/4$ " o.d. gauge 12 (B.W.G.)
and 16 in length

REFERENCES

[1] FONTANA, M.G. and GREEN, N.D.: "Connosion En-
gineering", McGraw-Hill Book Company, 1983.

[2] PARDUE, W.M.; BECK, F.H. and FONTANA, M.G.: Am.
Soc. Metals Trans. Quart., 54, pp 539-548(1961).

Recibido el 2 de febrero de 1987

 $-47-$